



Molecular Crystals and Liquid Crystals Incorporating Nonlinear Optics

Publication details, including instructions for authors and
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<http://www.tandfonline.com/loi/gmcl17>

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Version of record first published: 22 Sep 2006.

To cite this article: Shiro Kambe , Teruyuki Matsuoka , Maki Kawai , Tomoji Kawai & Makoto Takahasi
(1990): Effect of Elemental Substitution on the Superconductive Properties in $\text{Bi}_2\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_y$,
Molecular Crystals and Liquid Crystals Incorporating Nonlinear Optics, 184:1, 135-139

To link to this article: <http://dx.doi.org/10.1080/00268949008031751>

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EFFECT OF ELEMENTAL SUBSTITUTION ON THE SUPERCONDUCTIVE PROPERTIES IN $\text{Bi}_2\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_y$

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Abstract Effects of Pb substitution and N_2 annealing on the superconductivity of $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_y$ have been studied. Hole concentration of $\text{Bi}_2\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_y$ ($n=2,3$) was controlled by the substitution of Pb for Bi. The T_c dependence on the hole concentration in $n=3$ was different from that in $n=2$. The optimum hole concentration for highest T_c in $n=3$ phase was larger than that in $n=2$ phase. By annealing the $n=2$ phase in N_2 , $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{CaCu}_2\text{O}_y$ with $0.2 \leq x \leq 0.3$ became monoclinic. The T_c of this monoclinic $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{CaCu}_2\text{O}_y$ was observed above 97.5K.

INTRODUCTION

T_c 's of cuprate superconductors are very sensitive to their hole concentration (C_{hole})¹. Recently, Fukushima et al.³ have reported that Pb substitution for Bi in $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_y$ removes oxygen from the system with C_{hole} unchanged. On the contrary, some groups have revealed that Pb substitution produces a hole to $n=2$ phase^{2,4}. This discrepancy may be caused by the difference of the sintering condition particularly of partial O_2 pressure.

In this paper, Pb substitution effect on T_c was studied in $n=2$ and 3 phases. Especially for the $n=2$ phase prepared under various O_2 pressure, changes in the structure and the T_c dependence on Pb content was examined. It was found that 1) by sintering in $1/18 \text{ O}_2$, substitu-

tion of Pb for Bi produces a hole, 2) the optimum Chole per a unit cell for the maximum T_c in $n=3$ phase is larger than that in $n=2$ phase and that 3) Pb substitution or low oxygen pressure annealing promotes the structure change from pseudotetragonal into orthorhombic and for the Pb substituted ($x \geq 0.2$) system, N_2 annealing procedure leads the changes of the orthorhombic into the monoclinic structure. Maximum T_c observed for the monoclinic $n=2$ phase was above 97.5K.

EXPERIMENTAL

Samples were prepared by dissolving nitrates in water, followed by evaporation and calcination in air at 800°C for 2 hours. This powder was then reground, pressed into pellets, sintered in $1/18$ - $1/5$ atmosphere of O_2 for 40-60 hours at 800°C . They were post annealed in N_2 at 750°C for 8 hours. Four kinds of samples with different composition were synthesized, whose nominal ratios of Pb/Bi+Pb were (a)0, (b)0.05, (c)0.20 (d)0.30 and (e)0.50.

EPMA was carried out using AN 10/85S system by Link Analytical Co. attached to SEM (JSM-840A, JEOL) in order to determine the composition of the superconductor.

The structure of the compounds were determined by X-ray powder diffraction patterns using $\text{CuK}\alpha$ (RAD-IIIB, Rigaku denki). The peaks were assigned according to those indexed by M.Onoda et al⁹. Superconductive property was measured by the temperature dependence of both resistance and ac magnetic susceptibility. Resistivity measurements were carried out by a standard dc four probe method. Each sample was reground and pelletized prior to the ac susceptibility measurements. T_c was determined by the onset of as susceptibility.

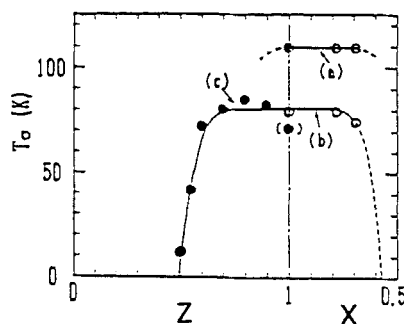


FIGURE 1 T_c dependence on Chole in $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{Ca}_{n(1-z)}\text{Y}_{nz}\text{Cu}_n\text{O}_y$

- a) hole doping by Pb substitution in $n=3$ phase
- b) hole doping by Pb substitution in $n=2$ phase
- c) hole quenching by Y substitution in $n=2$ phase¹

RESULTS AND DISCUSSION

T_c dependence on the hole concentration

As shown in Figure 1, T_c of $n=2$ phase decreased to 74.5K in $x=0.31$. This is due to the overdope of holes^{2,4} by Pb^{2+} substitution for Bi^{3+} . In $n=3$ phase, T_c dependence on C_{hole} was much smaller than that in $n=2$ phase. This suggests that the optimum C_{hole} for the highest T_c in $n=3$ phase is larger than that in $n=2$ phase. Further problem whether the optimum hole concentration per a Cu atom to give maximum T_c is different in $n=2$ and 3 has to be solved. It is necessary to measure how much holes are located in pyramidal and planar Cu-O layers in $n=3$ phase.

N₂ annealing effect

As shown in (a) and (b) of FIGURE 2, the structure of $Bi_{1.6}Pb_{0.4}Sr_2CaCu_2O_y$ changed drastically with reducing oxygen pressure. Reduction of oxygen pressure from 1/5 to 1/13 leads to the split of (2000,0200), which means that pseudotetragonal phase changed into orthorhombic³. By annealing sample (a) in N₂ at 750°C for 20 hours, (1130,1150,1170) peaks broadened as shown in (c). In particular, split of the peak was confirmed in (1170). This behavior cannot be explained by pseudotetragonal nor orthorhombic and by monoclinic structure. The peaks were indexed with monoclinic structure in $a=5.428\text{\AA}$, $b=5.358\text{\AA}$, $c=30.77\text{\AA}$ and $\gamma=90.8^\circ$. The peaks agreed well with calculated ones.

As shown in FIGURE 3, monoclinic $n=2$ phase was

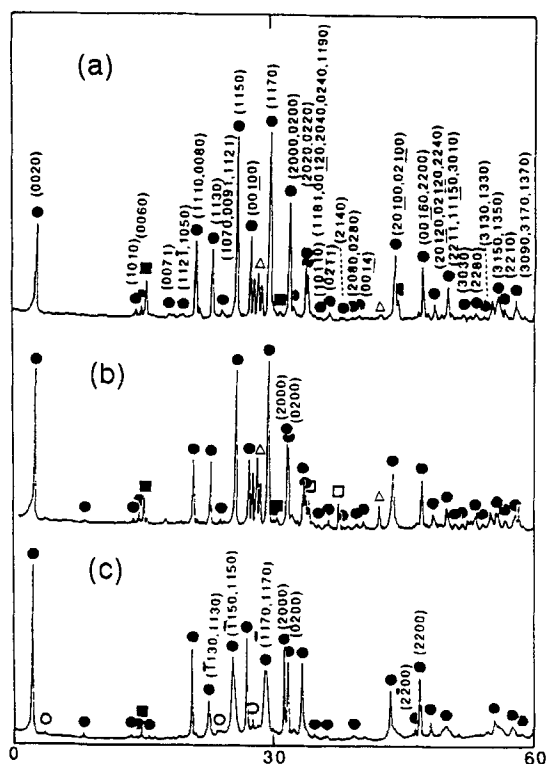


FIGURE 2 XRD patterns of $Bi_{1.6}Pb_{0.4}Sr_2CaCu_2O_y$

- (a) annealed in 1/5 O₂
 (b) annealed in 1/13 O₂
 (c) annealed in 1/5 O₂ followed by N₂

●: $n=2$ phase ○: $n=1$ phase ■: Ca PbO
 □: CuO △: unknown phase

prepared only by annealing in low oxygen pressure below 1/18 atm. with Pb of $0.2 \leq x \leq 0.3$. The c-axes length of monoclinic phases were 30.76–30.77 Å, which were shorter than that of orthorhombic or pseudotetragonal one (30.79–30.85 Å). It is reported that removing oxygen from n=2 leads to the increase of c-axis⁴, which is mainly considered to be due to the decrease of excess oxygen in Bi-O layer. The opposite behavior for monoclinic phase suggests the change of the structure in Bi-O layer. Another interesting feature is the increase in the intensity of (0020) peak with decreasing oxygen content. More precise analysis is now in progress.

The T_c of monoclinic n=2 phases were 97.5–98.0 K as shown in FIGURE 4(a). This temperature was extremely high in comparison with that before N₂ annealing (FIGURE 4(b)). This may be caused by the decrease of hole concentration. The optimum hole concentration to give the maximum T_c in n=2 phase is smaller than that of the

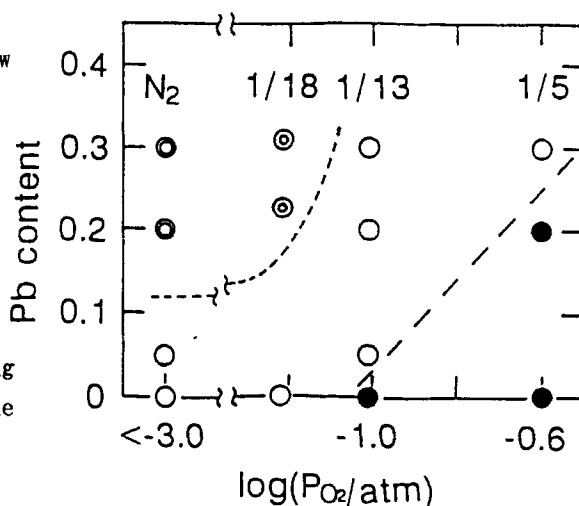


FIGURE 3 Phase diagram of $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{CaCu}_2\text{O}_y$

- ⊙ : monoclinic
- : orthorhombic
- : pseudotetragonal

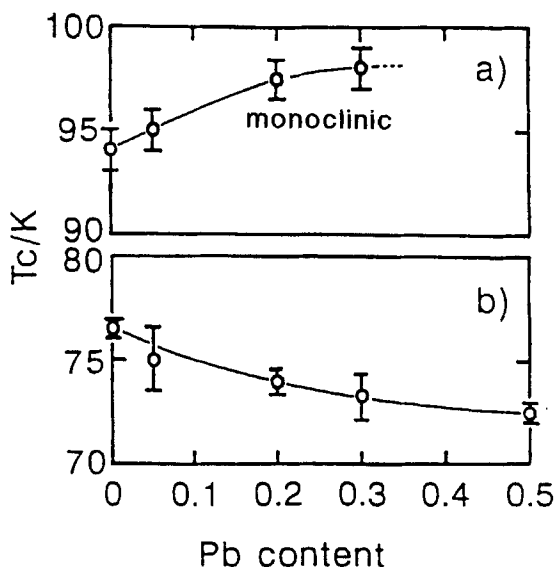


FIGURE 4 T_c dependence on Pb content

- (a) after N₂ annealing
- (b) before N₂ annealing

sample prepared in air^{2,5,6,7,8}. By reducing the system, the value of T_c is expected to increase. Now we are not sure whether the great increase in T_c up to 97.5K can be explained only by the reduction of Chole . Precise measurement on the hole concentration is now in progress.

CONCLUSION

Monoclinic $\text{Bi}_{2-2x}\text{Pb}_{2x}\text{Sr}_2\text{CaCu}_2\text{O}_y$ with $0.2 \leq x \leq 0.3$ was synthesized by N_2 annealing or by low O_2 pressure annealing. In $x=0.2$, lattice parameters of a , b , c and γ were 5.428Å, 5.358Å, 30.77Å and 90.8° , respectively. The observed T_c of monoclinic $n=2$ phase was above 97.5K, which was more than 20K higher than that before N_2 annealing.

ACKNOWLEDGMENTS

Authors are grateful to Dr.Y.Iimura for XRD measurement and to Dr.I.Higashi and Ms.R.Sekine for useful crystallographic discussion about the XRD data. A part of this work was carried out at the Frontier Research Program, the Institute of Physical and Chemical Research.

References

1. T.Tamegai, A.Watanabe, K.Koga, I.Oguro and Y.Iye, Jpn.J.Appl.Phys., **27**, L1074 (1988).
2. S.Kambe, T.Matsuoka, M.Kawai and M.Takahasi, submitted to Physica C.
3. N.Fukushima, H.Niu, S.Nakamura, S.Takano, M.Hayashi and K.Ando, Physica C, **49**, 777 (1989).
4. A.Maeda, M.Hase, I.Tsukada, K.Noda, S.Takebayashi and K.Uchinokura, submitted to Phys.Rev.B.
5. T.Tamegai, K.Koga, K.Suzuki, M.Ichihara, F.Sakai and Y.Iye, Jpn.J.Appl.Phys., **28**, L112 (1989).
6. J.Clayhold, S.J.Hagen, N.P.Ong, J.M.Tarascon and P.Barhous, Phys.Rev., **B39**, 7320 (1989).
7. D.E.Morris, C.T.Hultgren, A.M.Marletzt, J.Y.T.Wei, N.G.Asmar and J.H.Nickel, Phys.Rev.B, **39**, 6612 (1989).
8. A.Manthinam and J.B.Goodenough, Appl.Phys.Lett., **53**, 420 (1988).
9. M.Onoda, A.Yamamoto, E.Takayama-Muromachi and S.Takekawa, Jpn.J.Appl.Phys., **27**, L833 (1988).